

Interfacial Sites between Cobalt Nitride and Cobalt Act as Bifunctional Catalysts for Hydrogen Electrochemistry

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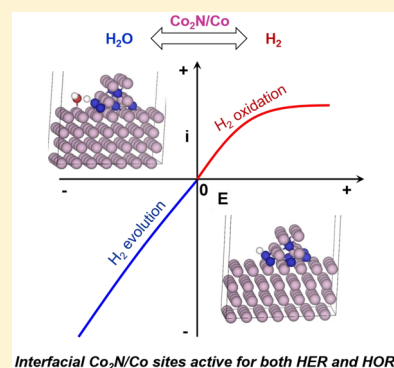
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Supporting Information

ABSTRACT: Hydrogen (H₂) electrochemistry primarily consists of two reactions: hydrogen evolution reaction in water for H₂ production (HER) and hydrogen oxidation reaction in hydrogen fuel cells for H₂ utilization (HOR). The realization of future hydrogen economy necessitates the development of low-cost and competent electrocatalysts for both HER and HOR. Herein, we report that partial nitridation of cobalt nanoparticles on current collectors results in rich Co₂N/Co interfacial sites, which exhibit bifunctional activity for hydrogen electrochemistry, rivaling the state-of-the-art Pt counterparts tested under similar conditions. Our combined experimental and theoretical computation results demonstrate that Co₂N/Co interfacial sites not only possess optimal hydrogen adsorption energy but also facilitate water adsorption and dissociation on the catalyst surface, all of which are beneficial to the electrocatalytic performance for both HER and HOR. In addition, our Co₂N/Co electrocatalysts also demonstrate great tolerance against CO poisoning during long-term H₂ oxidation.



Along with the increasing attention to exploring and utilizing sustainable energy sources, hydrogen (H₂) electrochemistry has attracted intense interest as H₂ production and oxidation play pivotal roles in the future hydrogen economy.^{1,2} In order to be economically attractive, low-cost and competent electrocatalysts are indispensable for both H₂ evolution and oxidation reactions (HER and HOR).^{3,4} Even though platinum-group metals, including Pt, Ir, Pd, and Rh, are still the state-of-the-art electrocatalysts of HER and HOR, their activities diminish quickly with along the increase of electrolyte pH.⁵ Consequently, there is a rise of research interest in developing earth-abundant electrocatalysts for hydrogen electrochemistry under neutral and alkaline conditions.^{6–8}

Because hydrogen adsorption energy (ΔG_{Had}) has been suggested as a common descriptor in predicting the activity of an electrocatalyst for HER and HOR, considerable efforts have been devoted to optimizing ΔG_{Had} through composition tuning,^{9–11} alloying,¹² surface modification,^{13–15} defect in-

duction,^{16,17} and molecular design.^{18,19} In neutral and alkaline electrolytes, water adsorption and dissociation have been recognized as critical steps in electrocatalytic H₂ evolution. Great success has been achieved in creating earth-abundant electrocatalysts for HER probably due to the enormous interest in water splitting during the past decade;^{20–22} however, relatively less progress has been achieved in HOR,^{23–26} even though HOR is a very important reaction for H₂ utilization in fuel cells. For example, a variety of metal nitride materials have been explored for electrocatalytic HER including Pt-decorated Ni₃N nanosheets,²⁷ Co/Ni₃N nanorods,²⁸ rodlike Co_xNi_yN,²⁹ Co_{5.47}N/N-doped carbon polyhedral,³⁰ graphdiyne-wrapped CoN_x nanosheets,³¹ CoN/Ni₃N nanowires,³² Co/CoN Janus nanoparticles embedded in N-doped carbon,³³ core-shell Co₂P-CoN nanoparticles in N-

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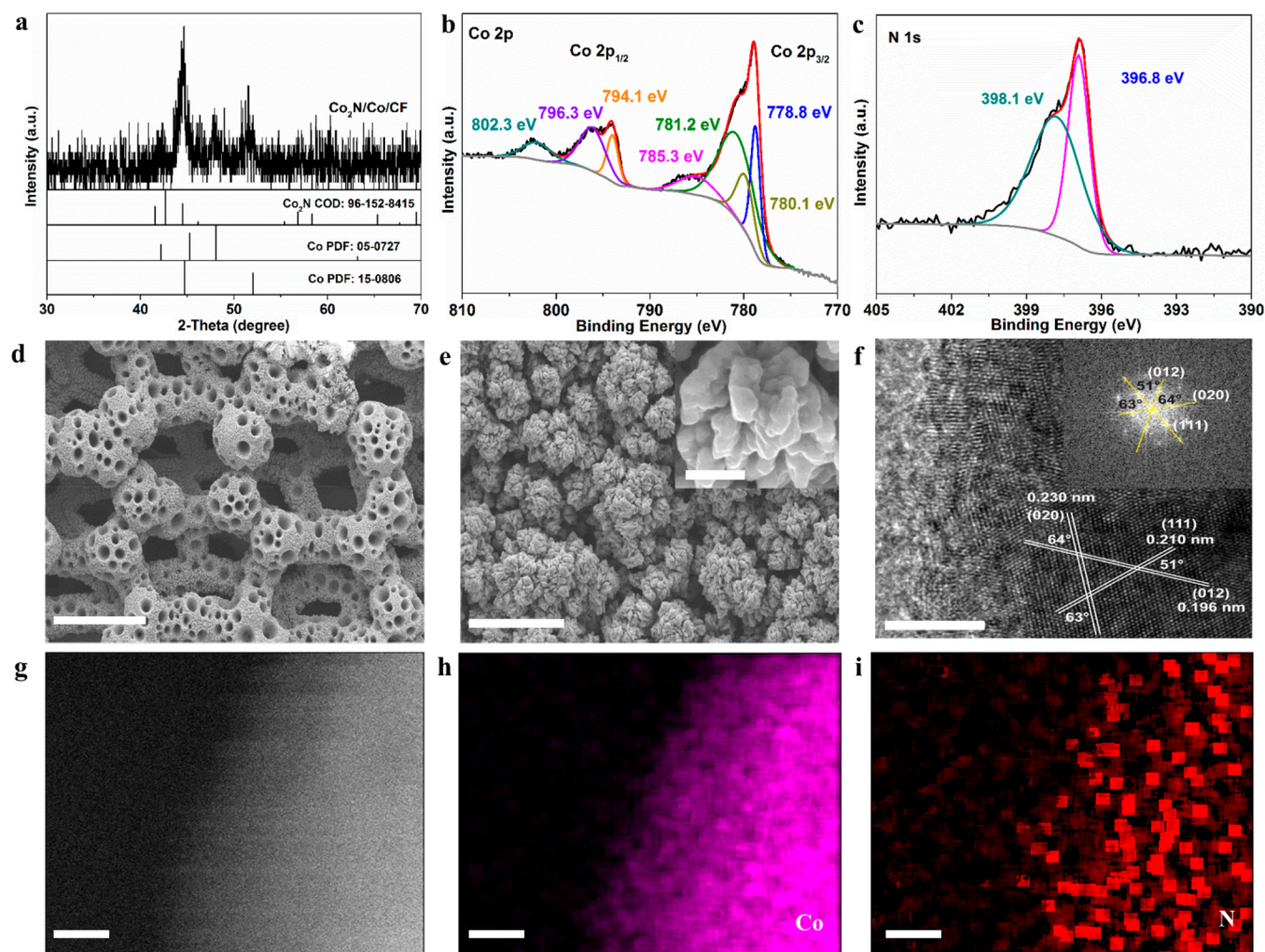


Figure 1. (a) XRD pattern of $\text{Co}_2\text{N}/\text{Co}/\text{CF}$ and the standard XRD patterns of Co_2N and Co (cubic and hexagonal phases) as reference. (b,c) High-resolution $\text{Co } 2p$ (b) and $\text{N } 1s$ (c) XPS spectra of $\text{Co}_2\text{N}/\text{Co}/\text{CF}$. (d,e) SEM images of $\text{Co}_2\text{N}/\text{Co}/\text{CF}$ at different magnifications. (f) HRTEM image showing the Co_2N component in $\text{Co}_2\text{N}/\text{Co}/\text{CF}$. The inset is a FFT pattern. (g–i) HAADF-STEM and corresponding elemental mapping images of Co and N in $\text{Co}_2\text{N}/\text{Co}/\text{CF}$. Scale bars: (d) $500 \mu\text{m}$; (e) $10 \mu\text{m}$ (500 nm in the inset); (f) 5 nm ; and (g–i) 2.5 nm .

doped carbon nanotubes,³⁴ and Co_4N nanowires on carbon cloth.³⁵ By creating heterostructures and tuning compositions, scientists have developed metal nitrides with much improved HER performance. However, most of the reported metal nitride electrocatalysts require energy-intensive hydrothermal treatment, expensive metal organic framework (MOF) templates, and complex organic polymerization, impeding their large-scale production. Moreover, those reported metal nitride electrocatalysts still need relatively large overpotentials ($68\text{--}190 \text{ mV}$) to deliver a HER benchmark current density of -10 mA cm^{-2} in neutral and alkaline electrolytes. Furthermore, the electrocatalytic HOR activity of metal nitrides has been barely investigated.^{23–25}

Recently, we reported a bifunctional electrocatalyst of $\text{Ni}_3\text{N}/\text{Ni}$ for both HER and HOR by creating interfacial sites between Ni_3N and Ni . Such an interfacing strategy shows great promise in developing competent hydrogen electrodes for water electrolyzers and hydrogen fuel cells.³⁶ However, it is unclear whether this strategy can be applied to construct other bifunctional transition metal-based electrocatalysts. Moreover, the relatively low hardness of nickel used in our previous work makes it difficult to be applied in producing structurally rigid

electrodes,³⁷ and the substrate nickel foam may also complicate the elucidation of the real performance of the active sites in $\text{Ni}_3\text{N}/\text{Ni}$. Hence, it is desirable to explore reactive inert, structurally flexible, and also low-cost electrode substrates with alternative electrocatalysts containing metal/compound interfaces.

Herein, we extend our interfacing strategy to the preparation of a bifunctional cobalt-based electrocatalyst and report that the interfacial sites between cobalt nitride and cobalt nanoparticles formed on cobalt foam ($\text{Co}_2\text{N}/\text{Co}/\text{CF}$) or carbon paper ($\text{Co}_2\text{N}/\text{Co}/\text{CP}$) exhibit superior bifunctional activity for hydrogen electrochemistry, rivaling that of Pt-based electrocatalysts evaluated under similar conditions. Our electrodeposition and nitridation steps can be carried out under atmospheric pressure in a facile and scalable manner, which is beneficial to the large-scale production of electrocatalysts. To the best of our knowledge, the resultant interfacial $\text{Co}_2\text{N}/\text{Co}$ electrocatalyst demonstrates extraordinary catalytic HER performance superior to that of nearly all reported cobalt-based electrocatalysts for both HER and HOR. Density functional theory (DFT) computational results support the fact that $\text{Co}_2\text{N}/\text{Co}$ interfacial sites bear suitable hydrogen and

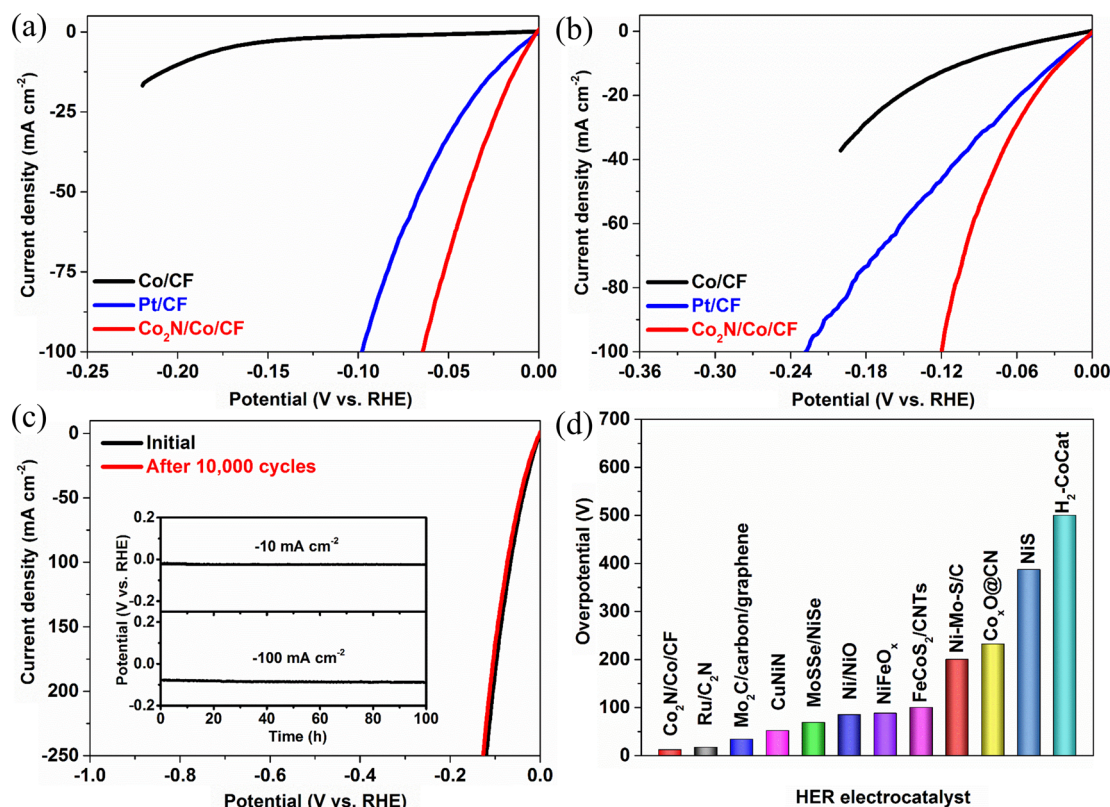


Figure 2. LSV curves of Co₂N/Co/CF, Pt/CF, and Co/CF for HER in 1.0 M KOH (a) and 1.0 M KPi (b). (c) LSV curves of Co₂N/Co/CF before and after 10 000 cycles at a scan rate of 100 mV s⁻¹ between 0 and -100 mV vs RHE in 1.0 M KOH. The inset shows the chronopotentiometry curves of Co₂N/Co/CF conducted at -10 and -100 mA cm⁻² without *iR* correction. (d) Comparison of overpotential requirements of selected HER electrocatalysts to deliver -10 mA cm⁻² for HER in 1.0 M KOH. All LSV curves were *iR*-corrected.

water adsorption energies and also facilitate water dissociation, all of which are believed to contribute to the overall activity of Co₂N/Co toward HER and HOR. Another exciting observation is that Co₂N/Co shows great tolerance to CO poisoning, which is widely known of plaguing Pt during long-term H₂ oxidation. The success of Co₂N/Co demonstrates an effective strategy of interfacing metals and metal compounds (e.g., metal nitrides) for enhanced hydrogen electrochemistry.

The synthesis of Co₂N/Co/CF started from the electro-deposition of highly porous cobalt nanoparticles on cobalt foam (Co/CF) followed by thermal nitridation in ammonia (see details in the Supporting Information). The X-ray diffraction pattern of Co₂N/Co/CF obtained from nitridation at 350 °C for 3 h is shown in Figure 1a (other samples are shown in Figures S1 and S2), indicating the presence of orthorhombic Co₂N, hexagonal Co, and cubic Co phases despite the background noise caused by the fluorescence interference of Co. The valence states of surface composition elements in Co₂N/Co/CF were probed via X-ray photoelectron spectroscopy (XPS). Figure 1b shows the high-resolution Co 2p XPS spectrum. The peaks at 780.1 and 796.3 eV can be assigned to Co–N, which are consistent with reported cobalt nitride.^{35,38} The broad peaks at 785.3 and 802.3 eV are satellite peaks. The peak at 781.2 eV is assigned to Co³⁺ due to surface oxidation upon exposure to air.³⁸ The peaks at 778.8 and 794.1 eV are ascribed to metallic Co 2p_{3/2} and 2p_{1/2}, respectively,³⁹ which is in agreement with those of pristine Co (Figure S3a,b). The Co 2p XPS results indicate the presence of both metallic cobalt and cobalt nitride in Co₂N/Co/CF. For the N 1s region (Figure 1c), the main peak

observed at 396.8 eV can be ascribed to the nitrogen species in Co₂N, while the shoulder peak at 398.1 eV likely results from the residual NH moieties of incomplete reaction with NH₃. The surface morphology of Co₂N/Co/CF was characterized by scanning electron microscopy (SEM) and compared to those of the parent Co/CF and pristine cobalt foam (CF). In contrast to the pristine CF with a smooth skeleton (Figure S4a,b), Co₂N/Co/CF displays a macroporous network structure of rough ligaments with numerous stacked particles over the skeleton surface (Figure 1d,e). Such a three-dimensional structure was mainly inherited from the parent Co/CF (Figure S4c,d). To further clearly identify the cobalt nitride phase among various polymorphs, a selected Co₂N crystal grain was characterized by high-resolution transmission electron microscopy (HRTEM). The well-resolved lattice fringes (Figure 1f) with interplanar spacing of 0.230, 0.210, and 0.196 nm measured through fast Fourier transform (FFT) correspond to the crystal planes of (020), (111), and (012) of Co₂N, respectively. The intersection angles between the two planes match very well with the theoretical values, which is direct evidence for the formation of orthorhombic Co₂N and also consistent with the XRD results.⁴⁰ To identify the Co₂N/Co interfacial sites, a HRTEM image of the Co₂N/Co interfacial region in Co₂N/Co/CF was collected and is shown in Figure S5a. Two kinds of spots in FFT of the selected area can be identified; one set includes larger spots, and another set has smaller dots (Figure S5b). The IFFT image from the selected larger spots (Figure S5c,d) indicates very crystalline metallic cobalt with its [111] zone-axis along the electron beam direction. However, the IFFT image from

the selected smaller dots (Figure S5e,f) clearly shows the formation of small Co₂N nanoparticles in a cobalt matrix with the zone-axis along [1–10]. In addition, scanning transmission electron microscopy (STEM) combined with energy-dispersive X-ray (EDX) mapping results obtained from the identical area is shown in Figures 1g–i and S6. Apparently, the nitrogen distribution in the composite is not uniform at the nanoscale, which indicates the presence of small nitride nanoparticles on the cobalt surface. EDX spectra from the selected areas (Figure S7) also show significantly different nitrogen contents in the different areas, even at the nanoscale. Collectively, these characterization results validate the formation of Co₂N on Co/CF, as well as Co₂N/Co interfacial sites in Co₂N/Co/CF.

Co₂N/Co/CF was first evaluated for its electrocatalytic H₂ evolution activity in both H₂-saturated alkaline (1.0 M KOH) and neutral (1.0 M potassium phosphate, KPi) electrolytes. All potentials reported herein are referenced to the reversible hydrogen electrode (RHE), and the reported current densities were normalized based on the geometric area of the electrode for the comparison of apparent activity. It should be noted that both nitridation temperature and duration were varied to achieve the optimized electrocatalyst, and it was found that nitridation at 350 °C for 3 h yielded a Co₂N/Co/CF sample with the best HER performance at both pH 7 and 14 (Figures S8–S11). Therefore, the Co₂N/Co/CF samples discussed in the following paragraphs were synthesized according to this optimized condition unless noted otherwise. For comparison, the pristine Co/CF and commercial Pt/C (20%) loaded on bare CF (Pt/CF) were also tested for catalytic HER under the same conditions. The loading mass of Pt/C was varied to achieve the best HER performance in both alkaline and neutral electrolytes (Figures S12 and S13). Figure 2a,b plots the linear sweep voltammetry (LSV) curves of Co₂N/Co/CF, Pt/CF, and Co/CF in 1.0 M KOH and KPi, respectively. In both alkaline and neutral electrolytes, Co₂N/Co/CF exhibited a catalytic HER takeoff at near-zero overpotential, similar to the benchmark Pt/CF. However, Co/CF required a much larger overpotential to initiate H₂ evolution. In 1.0 M KOH (Figure 2a), Co₂N/Co/CF was able to produce –10 and –100 mA/cm² at overpotential of merely 12 and 64 mV, respectively, substantially smaller than those required by Pt/CF (20 and 98 mV). At pH 7 (Figure 2b), Co₂N/Co/CF also exhibited excellent HER performance, demanding only 26 and 120 mV overpotentials to reach –10 and –100 mV/cm², which were apparently superior to Pt/CF tested under the same condition. The Tafel plots of Co₂N/Co/CF derived from their LSVs are shown in Figures S14 and S15. The Tafel slopes of Co₂N/Co/CF for HER in 1.0 M KOH and 1.0 M KPi were as low as 41.6 and 87.5 mV/decade, respectively, which are even smaller than those of Pt (73.3 and 108.5 mV/decade). These Tafel slopes of Co₂N/Co/CF indicate that the electrocatalytic HER kinetics likely follows the Heyrovsky mechanism (H₂O + M–H* + e[–] = M + H₂ + OH[–]), where H* represents a hydrogen atom adsorbed on a catalyst surface. The rate-limiting step of HER on Co₂N/Co/CF is closely related to the initial water dissociation and the subsequent H* adsorption.

The aforementioned HER performance based on the apparent current density could be overestimated for an electrocatalyst because of its highly porous structure. In order to make a fair comparison between Co₂N/Co/CF and the state-of-the-art Pt in electrocatalytic H₂ evolution, we further measured their respective electrochemically active surface area (ECSA) and normalized the obtained current

against the corresponding ECSA of each catalyst. It should be noted that all of the ECSA measurements were conducted in anhydrous acetonitrile with the aim of excluding the impact of hydration and protonation commonly observed for first-row transition metal compounds in aqueous media.⁴¹ Although the ECSA of Co₂N/Co/CF is 2.38 times lower than that of Pt/CF (2.5 mg cm^{–2}) (Figure S16a–d), the ECSA-normalized current densities of Co₂N/Co/CF for HER are still much higher than those of Pt/CF in both alkaline and neutral electrolytes (Figures S17 and S18). Figures S19a,b and S20a,b show the Nyquist plots with fitting curves and an equivalent circuit model and the Bode plots of Co₂N/Co/CF and Co/CF for HER at given potentials in 1.0 M KOH and KPi. The fitting results are listed in Table S1, showing the much smaller charge transfer resistance and fast kinetics of HER on Co₂N/Co/CF than those on Co/CF in both basic and neutral electrolytes.

To demonstrate that the excellent HER performance of Co₂N/Co is independent of the underlying current collector, such as the cobalt foam, we further synthesized Co₂N/Co on carbon paper (Co₂N/Co/CP) following the same synthetic procedure of Co₂N/Co/CF except for replacing CF with carbon paper. A suite of characterizations (XRD, SEM, and XPS) were also conducted on Co₂N/Co/CP (Figures S21, S22a,b, and S23a–d) to confirm the presence of Co₂N/Co with similar composition and morphology as those of Co₂N/Co/CF. Figure S24 presents the LSV and chronopotentiometry curves of Co₂N/Co/CP compared to those of Co₂N/Co/CF. The near overlap of these curves proves that the great HER performance of Co₂N/Co does not rely on a particular current collector; instead, it most likely originates from the real active sites located on Co₂N/Co.

Besides intrinsic activity, long-term stability is another important criterion in assessing any catalysts. The robustness of Co₂N/Co/CF for extended H₂ evolution was evaluated via repetitive cyclic voltammetry (CV) and chronopotentiometry. As depicted in Figure 2c, after 10 000 CV cycles, the HER overpotential of Co₂N/Co/CF required to achieve –100 mA cm^{–2} only increased by 7 mV in 1.0 M KOH. In addition, the stable chronopotentiometry curves collected at –10 and –100 mA/cm² for 100 h (Figure 2c inset) further corroborated its great robustness for long-term H₂ evolution. Similar stability was observed for Co₂N/Co/CF in neutral electrolyte (Figures S25 and S26). Postelectrolysis characterization revealed no apparent change in the morphology and composition of Co₂N/Co/CF (Figure S27a–d), implying its outstanding structural robustness and mechanical stability. Furthermore, the produced H₂ amount matched well the theoretically calculated quantity (Figure S28), which indicated that all of the passed charge was used to generate H₂, resulting in a nearly 100% Faradaic efficiency. Overall, to the best of our knowledge, Co₂N/Co/CF represents one of the most active HER electrocatalysts in alkaline and neutral electrolytes, surpassing a great number of reported nonprecious electrocatalysts (Figure 2d and Table S2).

Next, DFT computation was carried out to aid the understanding of the superior activity of Co₂N/Co/CF for H₂ production. We reasoned that a suitable model structure of Co₂N/Co is a few layers of Co₂N positioned on a Co basal plane. For a systematic comparison, three model systems with the lowest-energy surfaces of Co₂N(001), Co(001), and hybrid Co₂N/Co with a Co₂N(001) slab in contact with the Co(001) surface were considered. Due to the lack of free H⁺ in neutral and alkaline electrolytes, the commencement of electrocatalytic

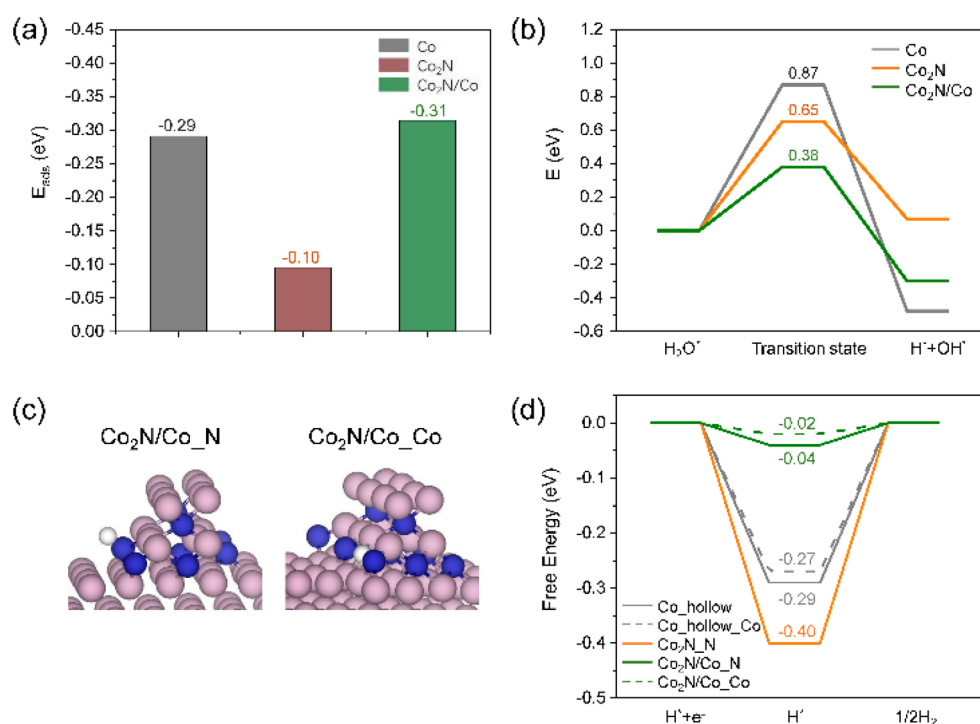


Figure 3. (a) Water adsorption energy on Co, Co₂N, and Co₂N/Co. (b) Reaction energetics for water dissociation on Co, Co₂N, and Co₂N/Co. (c) Optimized structures of H adsorption at two different positions of interfacial Co₂N/Co. Color code: pink (Co), blue (N), and white (H). (d) Free energy changes (ΔG_{Had}) of H adsorption on Co, Co₂N, and Co₂N/Co.

H₂ evolution at pH 7 and 14 demands water adsorption and dissociation on catalyst surface as the initial steps.⁴² The computed water adsorption energies on Co₂N/Co, Co₂N, and Co are compared in Figure 3a with their respective adsorption configurations shown in Figures S29a,b, S30a,b, and S31a,b, respectively. It is apparent that Co₂N/Co exhibits the strongest water adsorption among the three models, and the optimal water adsorption site is located at the interface between Co₂N and Co in Co₂N/Co with one of the H atoms in water approaching the edge N in Co₂N (Figure S32a,b), probably due to the hydrogen bond interaction. Such a configuration of water adsorption on Co₂N/Co is believed to be beneficial for its subsequent dissociation. Indeed, as shown in Figure 3b, the critical transition state of water dissociation has the lowest energy on Co₂N/Co. The detailed structure and energy of each state for water dissociation on Co₂N/Co, Co₂N, and Co can be found in the Supporting Information (Figures S32a,b, S33a,b, S34a,b, and S35a,b). A widely accepted consensus in hydrogen electrochemistry is that hydrogen adsorption free energy (ΔG_{Had}) on an electrocatalyst is a key descriptor of its activity.⁴³ A high catalytic performance is anticipated if ΔG_{Had} is close to 0 eV. Therefore, hydrogen adsorption on Co₂N/Co, Co₂N, and Co was also investigated computationally. The most plausible hydrogen adsorption sites on these models were considered (Figures S35–S44 and Table S3). The free energy changes of hydrogen adsorption at two interfacial sites of Co₂N/Co were calculated to be merely -0.02 and -0.04 eV, very close to the ideal value of 0 eV (Figure 3c,d). In contrast, both Co₂N and Co exhibit strong hydrogen affinity (-0.27 to -0.40 eV). Collectively, these DFT computational results demonstrate that the interfacial sites of Co₂N/Co can enhance water adsorption, facilitate water dissociation, and also exhibit optimal hydrogen adsorption energy, all of which contribute to its extraordinary HER activity.

Encouraged by the extremely small ΔG_{Had} value and nearly zero HER onset potential, we reasoned that Co₂N/Co could be a promising electrocatalyst for the reverse reaction of HER, the hydrogen oxidation reaction (HOR), as both HER and HOR share the same critical intermediate, adsorbed H (H_{ad}).^{4,44} The HOR performance of Co₂N/Co/CF was measured in H₂-saturated 0.1 M KOH electrolyte to mimic the alkaline condition of hydroxide exchange membrane fuel cells. Similar to HER, the best HOR activity was also obtained from Co₂N/Co/CF after nitridation at 350 °C for 3 h (Figures S45 and S46). Figure S47 compares the steady-state polarization curves of Co₂N/Co/CF collected in H₂- or Ar-saturated 0.1 M KOH in the same potential region of -0.06–0.1 V vs RHE. The apparent anodic current at positive potential (vs RHE) was obtained only in the presence of H₂, suggesting that H₂ oxidation takes place on Co₂N/Co/CF. Commercial Pt/C with optimized loading (1.5 mg cm⁻²) on CF (Pt/CF) was also included for comparison (Figure S48). As plotted in Figure 4a, the electrocatalytic HOR performance follows the order of Co₂N/Co/CF > Pt/CF > Co/CF. Co₂N/Co/CF showed a catalytic takeoff at 0 V vs RHE for HOR and produced higher current density than Pt/CF in the entire potential window (0–0.1 V vs RHE), even if the current density was normalized by their corresponding ECSA (Figure S49). In contrast, Co/CF displayed negligible catalytic activity for HOR, confirming that the enhanced HOR activity of Co₂N/Co/CF arises from the Co₂N/Co interfacial sites. In order to assess the intrinsic activity, the exchange current density was estimated from the micropolarization window from -10 to 10 mV vs RHE. Co₂N/Co/CF produced an exchange current density of -2.86 mA cm⁻², which was 1.4 and 28.5 times higher than that of Pt/CF and Co/CF, respectively (Figure S50). The stability of Co₂N/Co/CF for H₂ oxidation was further assessed by chronoamperometry at

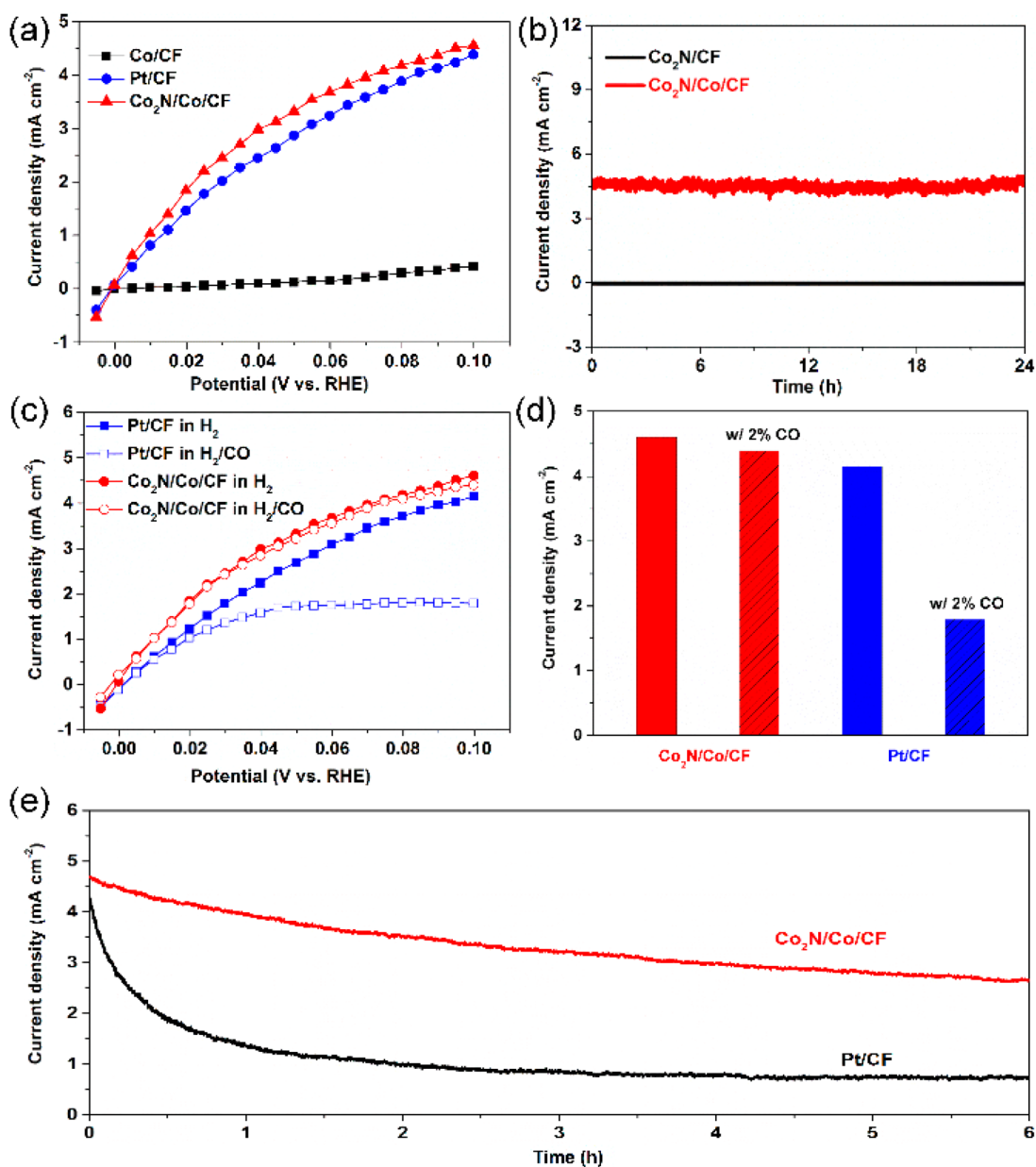


Figure 4. (a) Steady-state polarization curves of Co₂N/Co/CF, Co/CF, and Pt/CF for HOR in H₂-saturated 0.1 M KOH. (b) Chronoamperometry curves of Co₂N/Co/CF and Co/CF in H₂-saturated 0.1 M KOH measured at 0.09 V vs RHE. (c) Steady-state polarization curves and (d) comparison of the HOR performance of Co₂N/Co/CF and Pt/CF in 0.1 M KOH saturated with H₂ or H₂ with 2% (v/v) CO at 0.09 V vs RHE. (e) Chronoamperometry curves of Co₂N/Co/CF and Pt/CF in 0.1 M KOH saturated with H₂ with 2% (v/v) CO measured at 0.09 V vs RHE.

0.09 V vs RHE in 0.1 M KOH (Figure 4b), showing a stable current density of 4.5 mA cm⁻² in the presence of H₂, in striking contrast to the negligible capacitance current density obtained in the Ar-saturated KOH (Figure S51). Without nitridation, Co/CF only delivered no more than 0.1 mA cm⁻² in H₂-saturated 0.1 M KOH, in agreement with the crucial role of Co₂N/Co in catalyzing H₂ oxidation (Figure 4b). The excellent stability of Co₂N/Co/CF for H₂ oxidation was attributed to the robustness of Co₂N. Figures S52 and S53 present the XRD pattern and HRTEM image of Co₂N/Co/CF post HOR electrolysis, and the results indicate good retention of composition and retention of Co₂N/Co/CF after long-term electrolysis for H₂ oxidation in 0.1 M KOH. The XPS spectra of the postcatalysis Co₂N/Co/CF also exhibited little changes compared to that of pristine Co₂N/Co/CF (Figure S54).

Current industrial H₂ supply is mainly from steam reforming, producing cost-effective H₂ feeds yet with CO impurity, which is detrimental to Pt-based electrocatalysts.⁴⁵ Therefore, a high CO tolerance is desirable for HOR electrocatalysts. Herein, a CO tolerance test was also carried out for Co₂N/Co/CF and Pt/CF using a H₂ gas mixture with 2% CO (v/v). As shown in Figure 4c, the steady-state polarization curve of Co₂N/Co/CF showed no apparent degradation in the presence of CO, while Pt/CF indeed suffered from substantial CO poisoning. For instance, the anodic current density of Co₂N/Co/CF slightly decreased from 4.6 to 4.4 mA cm⁻² at 0.09 V vs RHE upon the addition of 2% CO in the H₂ feed (Figure 4d), whereas the current density of Pt/CF was dramatically reduced by 57%, decreasing from 4.15 to 1.79 mA cm⁻². Co₂N/Co/CF also exhibited a

much longer robustness than Pt/CF toward CO poisoning (Figure 4e). The anodic current density of Co₂N/Co/CF remained at ca. 3.5 mA cm⁻² after 6 h when electrolysis at 0.09 V vs RHE was carried out with a feed of 2% CO in H₂, whereas a current density of only 0.7 mA cm⁻² was produced on Pt/CF under the same condition.

In summary, we successfully constructed low-cost Co₂N/Co electrocatalysts through a facile electrodeposition and nitridation approach. The interfacial sites of Co₂N/Co exhibit excellent performance for both H₂ evolution and oxidation reactions, rivaling the activity of Pt-based electrocatalysts evaluated under similar conditions. Together with its remarkable CO tolerance, Co₂N/Co is a great catalyst candidate in the applications of water electrolyzers and alkaline hydrogen fuel cells. DFT calculations revealed the critical role of Co₂N/Co interfacial sites in promoting the adsorption and dissociation of water and optimizing the free energy of hydrogen adsorption. This work showcases that interface construction between metals and metal nitrides is an effective strategy in developing competent catalysts for the hydrogen electrochemistry.

■ ASSOCIATED CONTENT

Supporting Information

The Supporting Information is available free of charge on the ACS Publications website at DOI: 10.1021/acseenergylett.9b00738.

Details of the experimental methods, instruments, SEM, XRD, XPS, STEM-EDX, HRTEM, and electrochemical measurements (PDF)

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Notes

The authors declare no competing financial interest.

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